Improved mobility of zinc hexadecafluorophthalocyanine (ZnPcF₁₆) thin film transistors using *p*-6p as the inducing layer

YINGLI SUN, SHIRONG WANG^a, XIANGGAO LI^a, FENG MA^{*}

School of Chemistry and chemical Engineering, Tianjin University of Technology, Tianjin 300384 ^aSchool of Chemical Engineering and Technology, Tianjin University, Tianjin 300072, PR China

The n-type organic thin film transistors (OTFTs) employing zinc hexadecafluorophthalocyanine (ZnPcF₁₆) as the active layer and para-hexaphenyl (*p*-6p) as the inducing layer have been demonstrated. Compared with the ZnPcF₁₆-based OTFTs without the *p*-6p inducing layer, the charge carrier field-effect mobility (μ) of the ZnPcF₁₆/*p*-6p OTFTs is greatly improved. The mobility of the ZnPcF₁₆/*p*-6p OTFTs is 1.30×10⁻² cm²/V·s. The improved performance is attributed to the introduction of *p*-6p to form a highly oriented and continuous film of ZnPcF₁₆ with the molecular π - π stack direction parallel to the substrate.

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1. Introduction

Organic thin-film transistors (OTFTs) employing organic semiconductors as the active layer have received widely studies because of their potential application in displays, logic circuits, and sensors [1-4]. To date, the performance of p-type pentacene-based OTFTs has reached the level of a-Si devices [5]. Compared with p-type materials, the mobility of the n-type materials with high air stability is relatively low [6]. Therefore, how to improve the mobility of *n*-type materials becomes a challenge to meet the requirements of practical applications. Zinc hexadecafluorophthalocyanine (ZnPcF₁₆, shown in Fig. 1) is one of a few air-stable n-type organic semiconductors with a low electron field-effect mobility. To improve the performance of ZnPcF₁₆-based OTFTs, Bao et al. [7] have prepared transistors at different substrate deposition temperatures (T_d). The highest electron mobility of 1.2×10^{-3} cm^2/V s is obtained at a higher T_d of 215°C. Jiang *et al.* [8] have reported ZnPcF₁₆-based organic single-crystal field-effect transistors with a higher mobility about 1.1 cm²/V s. Highly oriented and continuous organic thin film is an ideal active layer to improve their device performance [9]. However, their thin films are normally composed of needle-like crystals with random orientations, which will induce more grain boundaries. The high defect concentration in grain boundaries of multi-crystal thin films usually leads to poor carrier transport, especially for phthalocyanine compounds [9-11].

In this study, we employ para-hexaphenyl (p-6p,

shown in Fig. 1) as the inducing layer to form well-organized $ZnPcF_{16}/p$ -6p thin films by physical vapor deposition technique. The morphology and structure of these thin films are examined by XRD and SEM. Furthermore, we fabricate $ZnPcF_{16}/SiO_2$ and $ZnPcF_{16}/p$ -6p OTFTs and compared their electrical characteristics to illustrate significant role of *p*-6p in the improvement of device mobility.



Fig. 1. Molecular structure of (a) $ZnPcF_{16}$ and (b) p-6p.

2. Experimental

The device configurations are given in Fig. 2. In the fabrication of the ZnPcF16/p-6p organic films and OTFTs, a heavily doped, n-channel Si wafer was used as the gate electrode and substrate, with a 300 nm thermally oxidized SiO_2 layer as the gate insulator. ZnPcF₁₆ and *p*-6p were purified twice by thermal gradient sublimation prior to processing. Firstly, p-6p was deposited on SiO₂ at a substrate temperature of 180 °C, carefully controlling the thickness 6 nm. Subsequently, a 7 nm ZnPcF₁₆ layer was deposited on the p-6p thin film. All organic films were deposited in vacuum $(10^{-4}-10^{-5} \text{ Pa})$ at a rate of 0.5 nm min⁻¹. In fabrication of the ZnPcF₁₆/p-6p OTFTs, the ZnPcF₁₆/p-6p films on the wafer substrate were rapidly moved to another vacuum chamber (10^{-4} Pa) for defining the source and drain electrodes by thermal evaporation of Au with a shadow mask. The width and length of the channel were 6000 µm and 200 µm, respectively. The ZnPcF₁₆/SiO₂ films and OTFTs were fabricated at the same conditions without introducing the p-6p organic layer. The transfer characteristics of the two transistors were measured with two Keithley 236 source-measurement units under ambient conditions at room temperature.



Fig. 2. The device configurations of (a) $ZnPcF_{16}/SiO_2$ and (b) $ZnPcF_{16}/p$ -6p thin film transistors

3. Results and discussion

Fig. 3 shows the XRD pattern of the ZnPcF₁₆/SiO₂ and ZnPcF₁₆/*p*-6p thin films. The X-ray diffraction was performed in a Rigaku D/max 2500 PC X-ray diffractometer with a Cu K α radiation ($\lambda = 1.54056$ Å). The diffraction peaks of the ZnPcF₁₆/SiO₂ film peak at $2\theta = 6.18$, corresponding to d_{200} spacing of 14.29 Å, is observed.



Fig. 3. X-ray diffraction spectra of $ZnPcF_{16}/SiO_2$ and $ZnPcF_{16}/p$ -6p thin films

The result indicates that ZnPcF₁₆ molecules are grown upright on the SiO₂ substrate with the metastable α -phase [7, 10, 12]. For ZnPcF₁₆/*p*-6p film, a sharp peaks at 2θ = 6.24, d_{200} = 14.15 Å, have a slight deviation from the diffraction peak of ZnPcF₁₆/SiO₂ thin film. It possibly originates from the difference between molecule-substrate interaction and intermolecular interaction [10]. These results also demonstrate that the ZnPcF₁₆ molecules are approximately standing-up on the *p*-6p layer.

Fig. 4 shows the SEM of ZnPcF₁₆/SiO₂ and ZnPcF₁₆/p-6p thin films. The SEM images were obtained by a FEI Nanosem 430 electron microscopy instrument. The surface image of ZnPcF₁₆/SiO₂ thin film is showed in Fig. 4a. The ZnPcF₁₆ molecules arrange on The FWHM of the ZnPcF₁₆/SiO₂ thin film is smaller than that of the $ZnPcF_{16}/p$ -6p thin film in Fig. 3. This means that $ZnPcF_{16}$ molecules were crystallized better in the ZnPcF₁₆/SiO₂ thin film. However, ZnPcF₁₆ molecules were randomly deposited on the amorphous SiO₂ substrate in the film-forming process. As shown in Fig. 4(a), $ZnPcF_{16}$ molecules arrange on the substrate disorderly, lacking continuous. The growth behavior of the p-6p thin film has been studied systematically [13]. A highly-ordered large-sized and smooth p-6p ultrathin film can supply a high quality substrate (the inducing layer) for the growth of phthalocyanine molecules. ZnPcF₁₆ molecules can be oriented after the employment of the *p*-6p inducing layer. The image of the $ZnPcF_{16}/p$ -6p film shows a strip grain pattern in Fig. 4(b). The π - π stack direction of ZnPcF₁₆ molecules in the $ZnPcF_{16}/p$ -6p thin film is parallel to the *p*-6p layer, which will facilitate the carrier transportation. Compared with the image of the ZnPcF₁₆/SiO₂ film, the ZnPcF₁₆/p-6p thin film shows higher order, fewer grain boundaries and traps in the ZnPcF₁₆ bulk. The highly oriented and continuous organic semiconductor thin film will lead to an improvement in the charge transport of organic electronic devices.



Fig. 4. SEM images of (a) $ZnPcF_{16}/SiO_2$ and (b) $ZnPcF_{16}/P-6p$ thin film

Fig. 5(a) and Fig. 5(b) show the typical transfer characteristics of the ZnPcF₁₆/SiO₂ and ZnPcF₁₆/*p*-6p OTFTs with different gate voltages at a fixed V_D of 100 V. The field effect mobility were extracted from figure Fig. 5a and Fig. 5b in the saturation region ($V \ge (V_G - V_T)$) based on

$$I_{DS} = \frac{W}{2L} \mu C_i (V_G - V_T)^2 \qquad (1)$$

Where I_{DS} is the drain-source current, W and L are the width and length of the channel, respectively, μ is the field-effect mobility, V_G is the gate voltage and V_T is the threshold Voltage. The capacitance per unit area of the insulator (C_i) is 8 nF/cm².



Fig. 5. Transfer curves characteristics of (a) $ZnPcF_{16}/SiO_2$ and (b) $ZnPcF_{16}/p$ -6p thin film transistors

From the transfer characteristics, we extract a mobility μ of 0.60×10^{-4} cm²/V s for ZnPcF₁₆/SiO₂ transistors while a mobility μ of 1.30×10^{-2} cm²/V s for ZnPcF₁₆/*p*-6p device. The mobility of ZnPcF₁₆/*p*-6p thin film transistor has been greatly improved due to the employment of the *p*-6p inducing layer leading to the formation of the high-quality ZnPcF₁₆ film.

4. Conclusion

In summary, highly oriented thin film of ZnPcF_{16} with the molecular π - π conjugated directions parallel to the substrate has been prepared. By applying this film-forming technique, the mobility of the n-channel ZnPcF_{16}/p -6p OTFTs is greatly improved. Employing *p*-6p as the inducing layer is an effective and simple method to fabricate high performance OTFTs for practical applications.

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^{*}Corresponding author: mafontune@sina.com.cn